

Large-Area MOCVD Growth of Ga₂O₃ in a Rotating Disc Reactor

NICK M. SBROCKEY,^{1,3} THOMAS SALAGAJ,¹ ELANE COLEMAN,¹
GARY S. TOMPA,¹ YOUNGBOO MOON,² and MYUNG SIK KIM²

1.—Structured Materials Industries, Inc., 201 Circle Drive North, Unit #102, Piscataway, NJ 08854, USA. 2.—UJL, A1513 Kwangkyo-ro 145, Yeongtong-gu, Suwon-si, Gyenggi-do 443270, Korea. 3.—e-mail: sbrockey@structuredmaterials.com

Gallium oxide is a wide-bandgap semiconductor material which is being developed for a range of electronic and electrooptic device applications. Commercial implementation of these devices will require production-scale technology for Ga₂O₃ film deposition. This work demonstrated deposition of uniform Ga₂O₃ films on both 50-mm-diameter (0001) sapphire and 200-mm-diameter (100) silicon substrates, using a rotating disc metalorganic chemical vapor deposition reactor. The source reactants were trimethylgallium and oxygen. The resulting Ga₂O₃ films were smooth, optically transparent, and highly insulating and had excellent thickness uniformity. Ga₂O₃ films deposited on (0001) sapphire at temperatures of at least 600°C and pressures of at least 45 Torr consisted of ($\bar{2}01$)-oriented β -Ga₂O₃ in the as-deposited state. The β -crystal structure was shown to be stable on annealing to 800°C, in either air or nitrogen atmosphere. A Ga₂O₃ film deposited at a lower temperature was shown to crystallize to a similar ($\bar{2}01$)-oriented β -Ga₂O₃ structure on postdeposition annealing at 800°C.

Key words: Ga₂O₃, thin films, MOCVD, epitaxy

INTRODUCTION

Recently, there has been significant interest in the semiconductor material Ga₂O₃ for a wide range of electronic and electrooptic device applications. Ga₂O₃ has a very attractive combination of properties, including a wide bandgap with measured values ranging from 4.4 eV to 4.9 eV,^{1–3} a high breakdown field of 2.80 MV/cm to 5.70 MV/cm,^{4,5} and a dielectric constant of 10.⁴ In addition, high-quality β -Ga₂O₃ substrates can be produced by standard techniques,⁶ and are now commercially available. Ga₂O₃ is being developed for a range of applications, including power semiconductor devices,⁷ gas sensors,^{8,9} transparent conductive oxides,² ultraviolet (UV) photodetectors,^{3,10,11} and as gate dielectric layers for GaAs^{4,12} and GaN⁵ devices. Ga₂O₃ films could also be used as buffer layers for subsequent GaN film deposition.^{13,14} These applications will require high-quality

Ga₂O₃ films to be deposited on Ga₂O₃ or other substrates. Many techniques have been investigated for deposition of Ga₂O₃ films, including evaporation,^{4,15} the sol-gel method,³ chemical solution deposition,⁹ spray pyrolysis,^{16,17} radio frequency (RF) sputtering,^{8,18} pulsed laser deposition,² atomic layer deposition,¹⁹ molecular beam epitaxy (MBE),^{7,12,20,21} and metalorganic chemical vapor deposition (MOCVD).^{10,11,22–29}

Unlike competing wide-bandgap materials such as SiC, GaN, and diamond, Ga₂O₃ single crystals can be grown from the melt.⁶ Large-area Ga₂O₃ substrates thus have the potential to become more economical and more readily available than any of these competing materials. Therefore, it is important to develop large-area film deposition technology for Ga₂O₃ films for use in wide-bandgap device applications. Deposition of Ga₂O₃ films by MOCVD offers many advantages for ultimate device fabrication, including conformal deposition over device topography and the capability for scale-up to high-volume production. Among the published studies of

(Received August 8, 2014; accepted November 28, 2014)

Ga_2O_3 film deposition by MOCVD, only one other research group^{11,26} specifically identified the deposition system as one capable of scaling to large-area wafers. The rotating disc MOCVD reactor is a proven technology for achieving uniform film deposition on large-area substrates.^{30,31} This work aimed to demonstrate large-area growth of device-quality $\beta\text{-Ga}_2\text{O}_3$ films using rotating disc MOCVD reactor technology.

EXPERIMENTAL PROCEDURES

All Ga_2O_3 film depositions were done in a rotating disc MOCVD reactor. The reactor has a 16-inch-diameter stainless-steel chamber. The gas flow geometry is vertical, from an upper showerhead. The substrate platter is a horizontal, high-speed rotating disc with diameter of 31.75 cm. The platter can be configured for single-wafer (200 mm or 300 mm) or multiple-wafer (150 mm or smaller) loads. The MOCVD reactants were trimethylgallium (TMGa) and oxygen. Argon was used as the carrier gas. The deposition temperature range investigated was 500°C to 650°C. The deposition pressure range was 4 Torr to 65 Torr. The O_2 flow rates were 750 standard cm^3/min (sccm) to 1800 sccm. The substrates used for the majority of the deposition trials were 50-mm-diameter (0001) sapphire wafers. Additional deposition trials were also done on 200-mm (100) silicon wafers.

For Ga_2O_3 films deposited on either (100) silicon or (0001) sapphire, thickness was determined by optical reflectance spectroscopy using white light. For Ga_2O_3 films on (100) silicon, no other characterization was done since the intention in this case was to demonstrate uniform deposition on large-area substrates. Ga_2O_3 films on (0001) sapphire were also characterized by x-ray diffraction for crystal structure and orientation, optical microscopy for general film morphology, and four-point probe for sheet resistivity. X-ray diffraction was done using Cu K_α radiation. Postdeposition annealing was done in a conventional tube furnace under flowing nitrogen or in a conventional box furnace in static air. All annealing trials were done at 800°C for 15 min, followed by furnace cool down to near room temperature.

RESULTS

The measured thickness of the MOCVD-deposited Ga_2O_3 films was generally in the range of 100 nm to 320 nm. All films were smooth and featureless as observed by optical microscopy up to 1500 \times magnification. All Ga_2O_3 films were insulating, with sheet resistivity beyond the measurement range of our equipment (10^{10} Ohms/square). The Ga_2O_3 films on sapphire substrates were visibly transparent. Results of initial MOCVD process optimization are summarized in Figs. 1, 2, and 3, which show the measured Ga_2O_3 film growth rate as a function of

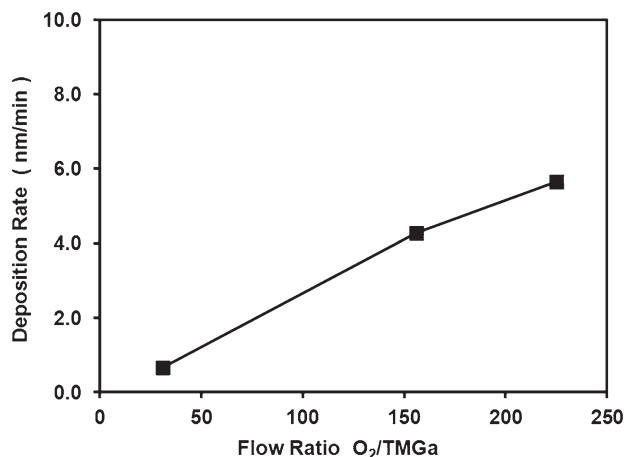


Fig. 1. Measured growth rate versus O_2/TMGa flow ratio for Ga_2O_3 films deposited on (0001) sapphire at 600°C and 15 Torr.

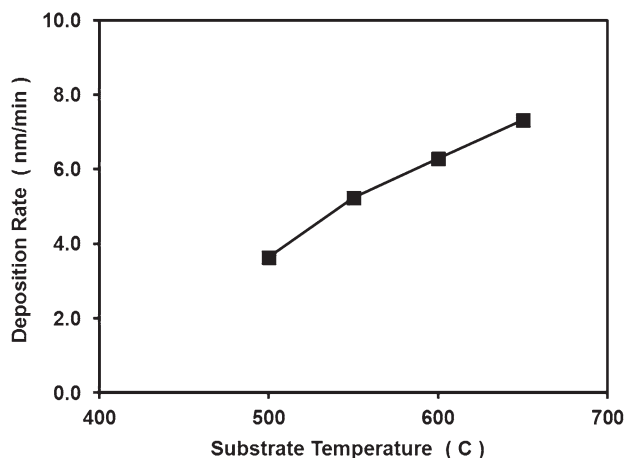


Fig. 2. Measured growth rate versus deposition temperature for Ga_2O_3 films deposited on (0001) sapphire at 45 Torr with O_2/TMGa flow ratio of 225.

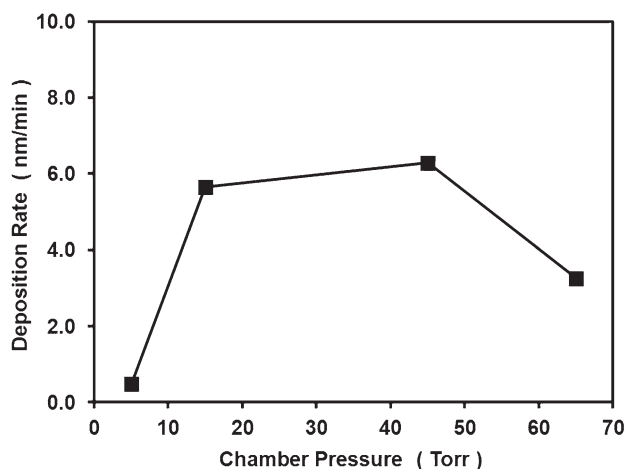


Fig. 3. Measured growth rate versus deposition pressure for Ga_2O_3 films deposited on (0001) sapphire at 600°C with O_2/TMGa flow ratio of 225.

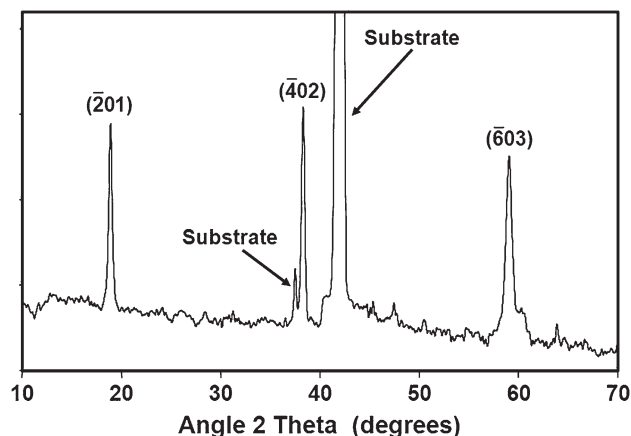


Fig. 4. X-ray diffraction results for Ga_2O_3 film deposited on (0001) sapphire at 600°C, 45 Torr, and O_2/TMGa flow ratio of 225.

the O_2/TMGa flow ratio, deposition temperature, and deposition pressure, respectively.

X-ray diffraction results for Ga_2O_3 films deposited at temperature of at least 600°C and pressure of at least 45 Torr indicated a $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ structure, as shown in Fig. 4. All films deposited at lower temperature or pressure showed no x-ray diffraction peaks in the as-deposited condition. Postdeposition annealing was done on a Ga_2O_3 film deposited at temperature of 550°C, pressure of 45 Torr, and O_2/TMGa flow ratio of 225. This sample originally showed no x-ray diffraction peaks in the as-deposited condition. Annealing for 15 min at 800°C in air resulted in crystallization to a $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ structure, with an x-ray diffraction pattern similar to that shown in Fig. 4. Two additional film samples deposited at 600°C, 45 Torr, and O_2/TMGa flow ratio of 225 were also postdeposition annealed in flowing nitrogen and in static air. X-ray diffraction analysis of these samples showed the $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ structure in the as-deposited state. Each sample was annealed for 15 min at 800°C in the specified atmosphere. For both samples, the x-ray diffraction pattern was essentially unchanged, confirming the stability of the β -crystal structure on annealing to 800°C in both air and nitrogen.

Figure 5 shows a photograph of a 200-mm (100) silicon wafer with MOCVD-deposited Ga_2O_3 film of 90 nm nominal thickness. The insert in Fig. 5 shows a 50-mm single-side-polished (0001) sapphire substrate with an MOCVD-deposited Ga_2O_3 film of nominal 320 nm thickness. Figure 6 shows the measured thickness of the Ga_2O_3 film on (100) silicon, as a function of position across the wafer. From the data of Fig. 6, the thickness uniformity for the Ga_2O_3 film is 3.3% across the 200-mm wafer.

DISCUSSION

From the process optimization results, the best MOCVD process conditions for achieving high

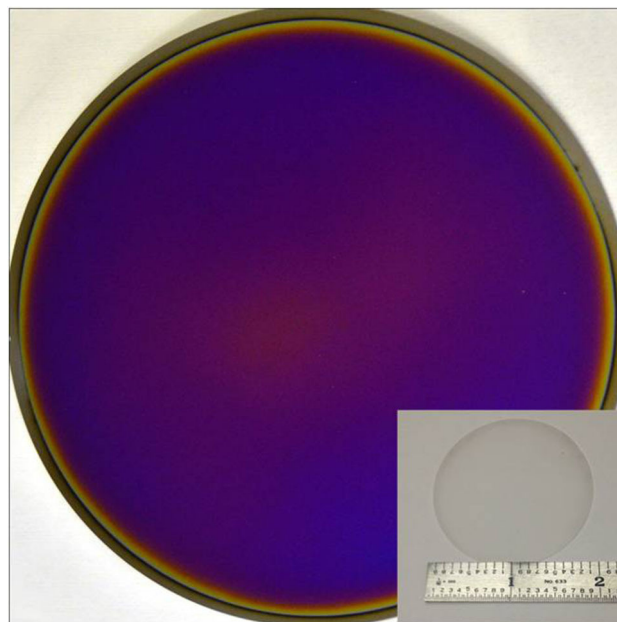


Fig. 5. Photograph of a 200-mm (100) silicon wafer with MOCVD-deposited Ga_2O_3 film of nominal 90 nm thickness. The insert shows an MOCVD-deposited Ga_2O_3 film of nominal 320 nm thickness, on a 50-mm single-side-polished (0001) sapphire wafer. The size scale in the insert is applicable for both images.

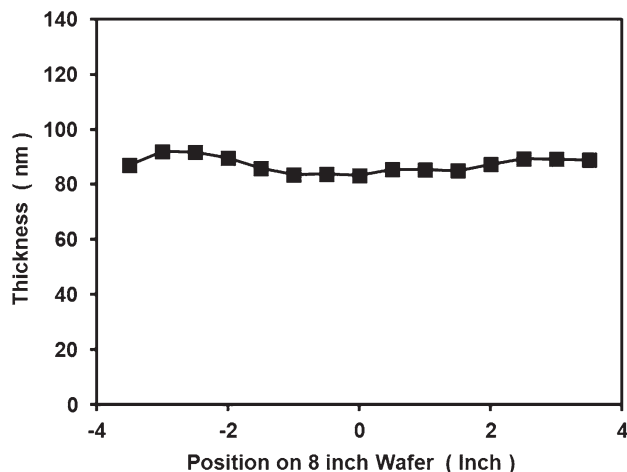


Fig. 6. Measured thickness of Ga_2O_3 film deposited on a 200-mm (100) silicon wafer, as a function of position across the wafer.

deposition rates and highly $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ are pressure of 45 Torr to 65 Torr, temperature of at least 600°C, and O_2/TMGa flow ratio of 225. These optimum conditions are slightly different from those reported in other studies of Ga_2O_3 MOCVD,^{11,26} which used lower O_2 flow rates. The higher O_2 flow rates used in this study are better suited to large chamber sizes for large-area Ga_2O_3 film deposition.

The $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ structure has also been observed in several other studies of Ga_2O_3 films deposited on (0001) sapphire substrates by MOCVD^{10,22–26,28} and by MBE.²⁰ The x-ray diffraction

technique used in this work cannot confirm epitaxy of the Ga_2O_3 film, since it does not measure rotational orientation within the film plane. However, other works have confirmed epitaxial $(\bar{2}01)$ -oriented $\beta\text{-Ga}_2\text{O}_3$ films on (0001) sapphire through pole figure measurements¹⁵ and by transmission electron microscopy.²⁸ Analysis using a crystal model showed the [201] direction of the $\beta\text{-Ga}_2\text{O}_3$ to be perpendicular to the sapphire {100} planes.¹² In light of these works, it is likely that the $\beta\text{-Ga}_2\text{O}_3$ films deposited on (0001) sapphire by MOCVD in this work are in fact epitaxial.

Postdeposition annealing has been shown to have a significant effect on the electrical and optical properties of $\beta\text{-Ga}_2\text{O}_3$ films on (0001) sapphire, as characterized by Hall measurements,¹¹ photoconductivity,¹¹ cathodoluminescence,¹¹ current–voltage (I – V) characteristics,^{25,26} and photoluminescence.²⁶ Therefore, it is important to characterize the annealing behavior of the Ga_2O_3 films, in order to achieve optimum properties for the ultimate device application. The Ga_2O_3 films produced in this study were shown to maintain the β -crystal structure on annealing up to 800°C, in agreement with other annealing studies.^{23,25} We did not observe any transformation to the α -structure, as has been reported by others.²⁶ The annealing atmosphere (air versus nitrogen) was shown to have no effect on the film quality as measured by x-ray diffraction in this study.

In summary, MOCVD in a rotating disc reactor was shown to be a production-worthy technique to produce Ga_2O_3 films for device applications. Deposition rates up to 7.3 nm/min were demonstrated for $\beta\text{-Ga}_2\text{O}_3$ films on (0001) sapphire substrates. We anticipate higher deposition rates can be achieved with higher precursor delivery rates. A thickness uniformity of 3.3% was demonstrated for an MOCVD-deposited Ga_2O_3 film on a 200-mm silicon wafer. Epitaxial $(\bar{2}01)$ $\beta\text{-Ga}_2\text{O}_3$ films on (0001) sapphire were achieved by deposition at temperatures of at least 600°C and pressures of at least 45 Torr. Epitaxial $(\bar{2}01)$ $\beta\text{-Ga}_2\text{O}_3$ films can also be achieved by postdeposition annealing of films deposited at lower temperatures and pressures. The epitaxial $(\bar{2}01)$ $\beta\text{-Ga}_2\text{O}_3$ film structure was shown to be stable on annealing up to 800°C, as observed by x-ray diffraction.

REFERENCES

1. H.H. Tippins, *Phys. Rev.* 140, 316 (1965).
2. M. Orita, H. Ohta, M. Hirano, and H. Hosono, *Appl. Phys. Lett.* 77, 4166 (2000).

3. Y. Kokubun, K. Miura, F. Endo, and S. Nakagomi, *Appl. Phys. Lett.* 90, 031912 (2007).
4. M. Passlack, E.F. Schubert, W.S. Hobson, M. Hong, N. Moriya, S.N.G. Chu, K. Konstadinidis, J.P. Mannaerts, M.L. Schnoes, and G.J. Zydzik, *J. Appl. Phys.* 77, 686 (1995).
5. C.-T. Lee, H.-W. Chen, and H.-Y. Lee, *Appl. Phys. Lett.* 82, 4304 (2003).
6. J. Zhang, B. Li, C. Xia, G. Pei, Q. Deng, Z. Yang, W. Xu, H. Shi, F. Wu, Y. Wu, and J. Xu, *J. Phys. Chem. Solids* 67, 2448 (2006).
7. M. Higashiwaki, K. Sasaki, A. Kuramata, T. Masui, and S. Yamakoshi, *Appl. Phys. Lett.* 100, 013504 (2012).
8. C.-I. Baban, Y. Toyoda, and M. Ogita, *J. Appl. Phys.* 43, 7213 (2004).
9. M. Bartic, M. Ogita, M. Isai, C.-L. Baban, and H. Suzuki, *J. Appl. Phys.* 102, 023709 (2007).
10. D.-S. Wu, S.-L. Ou, R.-H. Horng, P. Ravadgar, T.-Y. Wang, and H.-Y. Lee, *Proc. SPIE* 8263 (2012).
11. C.-Y. Huang, R.-H. Horng, D.-S. Wu, L.-W. Tu, and H.-S. Kao, *Appl. Phys. Lett.* 102, 011119 (2013).
12. W. Priyantha, G. Radhakrishnan, R. Droopad, and M. Passlack, *J. Cryst. Growth* 323, 103 (2011).
13. X. Zi-Li, Z. Rong, X. Chang-Tai, X. Xiang-Qian, H. Ping, L. Bin, Z. Hong, J. Ruo-Lian, S. Yi, and Z. You-Dou, *Chin. Phys. Lett.* 25, 2185 (2008).
14. S. Ito, K. Takeda, K. Nagata, H. Aoshima, K. Takehara, M. Iwaya, T. Takeuchi, S. Kamiyama, I. Akasaki, and H. Amano, *Phys. Status Solidi* 9, 519 (2012).
15. S. Nakagomi and Y. Kokubun, *J. Cryst. Growth* 349, 12 (2012).
16. T. Kawaharamura, G.T. Dang, and M. Furuta, *J. Appl. Phys.* 51, 040207 (2012).
17. S.-D. Lee, K. Akaiwa, and S. Fujita, *Phys. Status Solidi C* 10, 1592 (2013).
18. Y. Wei, Y. Jinliang, W. Jiangyan, and Z. Liying, *J. Semicond.* 33, 073003 (2012).
19. F.K. Shan, G.X. Liu, W.J. Lee, G.H. Lee, I.S. Kim, and B.C. Shin, *J. Appl. Phys.* 98, 023504 (2005).
20. E.G. Villora, K. Shimamura, K. Kitamura, and K. Aoki, *Appl. Phys. Lett.* 88, 031105 (2006).
21. M. Higashiwaki, K. Sasaki, T. Kamimura, M.H. Wong, D. Krishnamurthy, A. Kuramata, and T. Masui, *Appl. Phys. Lett.* 103, 123511 (2013).
22. D. Gogova, G. Wagner, M. Baldini, M. Schmidbauer, K. Irmischer, R. Schewski, Z. Galazka, M. Albrecht, and R. Fornari, *J. Cryst. Growth* 401, 665 (2013).
23. P. Ravadgar, R.-H. Horng, S.-D. Yao, H.-Y. Lee, B.-R. Wu, S.-L. Ou, and L.-W. Tu, *Opt. Express* 21, 24599 (2013).
24. P. Ravadgar, R.-H. Horng, L.-W. Tu, S.-L. Ou, H.-P. Pan, and S.-D. Yao, *Proc. SPIE* 8626 (2013).
25. R.-H. Horng, P. Ravadgar, *Proc. SPIE* 8626 (2013).
26. P. Ravadgar, R.H. Horng, and T.Y. Wang, *ECS J Solid State Sci. Technol.* 1, N58 (2012).
27. W. Mi, J. Ma, Z. Zhu, and C. Luan, *J. Cryst. Growth* 354, 93 (2012).
28. Y. Lv, J. Ma, W. Mi, C. Luan, Z. Zhu, and H. Xiao, *Vacuum* 86, 1850 (2012).
29. H.W. Kim and N.H. Kim, *Appl. Surf. Sci.* 230, 301 (2004).
30. William G. Breiland and Greg H. Evans, *J. Electrochem. Soc.* 138, 1806 (1991).
31. G.S. Tompa, P.A. Zawadzki, K. Moy, M. McKee, A.G. Thompson, A.I. Gurary, E. Wolak, P. Esherick, W.G. Breiland, G.H. Evans, N. Bulitka, J. Hennessy, and C.J.L. Moore, *J. Cryst. Growth* 145, 655 (1994).